

Introduction

For an ongoing project to study the transport processes within a volcanic conduit, including crystallization and degassing, we are coupling a thermodynamics model (TD) to a computational fluid dynamics (CFD) model allowing for a fully thermo- and fluid dynamically consistent simulation.

Despite the wide range of TD model families available, e.g. MELTS (Ghiorso and Sack, 1995), THER-MOCALC (Holland and Powell, 2011), PERPLE_X (Connolly, 2005), the need to re-implement a TD model to suit our cause is become apparent for the reasons stated below. Here, we can only give a short account of our approach to one of the problems encountered: The characterization of miscibility gaps.

Why re-invent the wheel?

Given all the established thermodynamics software and toolboxes, we still decided to endeavour reimplementing one of them for two reasons:

Ensuring computing performance

The interface between separate TD and CFD programs forms a severe bottleneck. Direct integration and compilation of TD into the CFD code is necessary for acceptable run-times.

Learning how to use TD model properly Using a complex TD model requires knowledge of its range of applicability to stay within it, or to know when it is safe to transgress those limits (e.g. a mixing-model being valid for what bulk compositions). This information is often not obvious at first glance.

On gaining more insight, a third requirement was added to the list and soon became paramount:

Ensuring numerical consistency

Fully coupling a TD and a CFD model creates a feedback loop where the output of one model at one time-step is the other's input at the next and vice versa. This amplifies inconsistencies resulting solely from the *numerical* implementation of a *physically* consistent TD model that would be insignificant under normal operation.

We chose to re-implement the TD model of Holland and Powell (2011) using their database and mixing models but writing a new solver in Fortran to be compatible with our CFD program (Flow3D).

Figure 1: Topography of the Gibbs energy of mixing G_{mix} (eq. 1, on the right) for a fictive ternary system $Al^{I}Al^{II}O_{3}$, $Mg^{I}Si^{II}O_{3}$, $Fe^{I}Si^{II}O_{3}$ with mixing on two but indistinguishable sites. It shows three local maxima, already suggesting miscibility gaps, and three local minima close to the three end-members.

Towards a self-consistent thermodynamic magmatic model for conduit transport processes

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The thermodynamics model

TD models solve a system's equation of state (EOS) given as a thermodynamic potential, e.g. the Gibbs free energy G, at a given state (pressure, temperature, bulk composition, PTX) under the assumption of equilibrium for the stable phase composition. They consist of three parts that are, at least in principle, independent from each other:

- The so-called *database*, i.e. the EOSs of the pure end-member phases,
- the mixing models (also activity-composition) of how the singles phases interact, and
- a solver to find the composition at equilibrium under given PTX conditions.

Parameters: T = 1474 K; $G_i = 0$, $\delta G_i = 8.495$, $W_{AM} = 8$, $W_{AF} = 6$, $W_{MF} = 4 \text{ kJ/mol}$; $a_i = [1.0, 1.1, 1.4]$, and

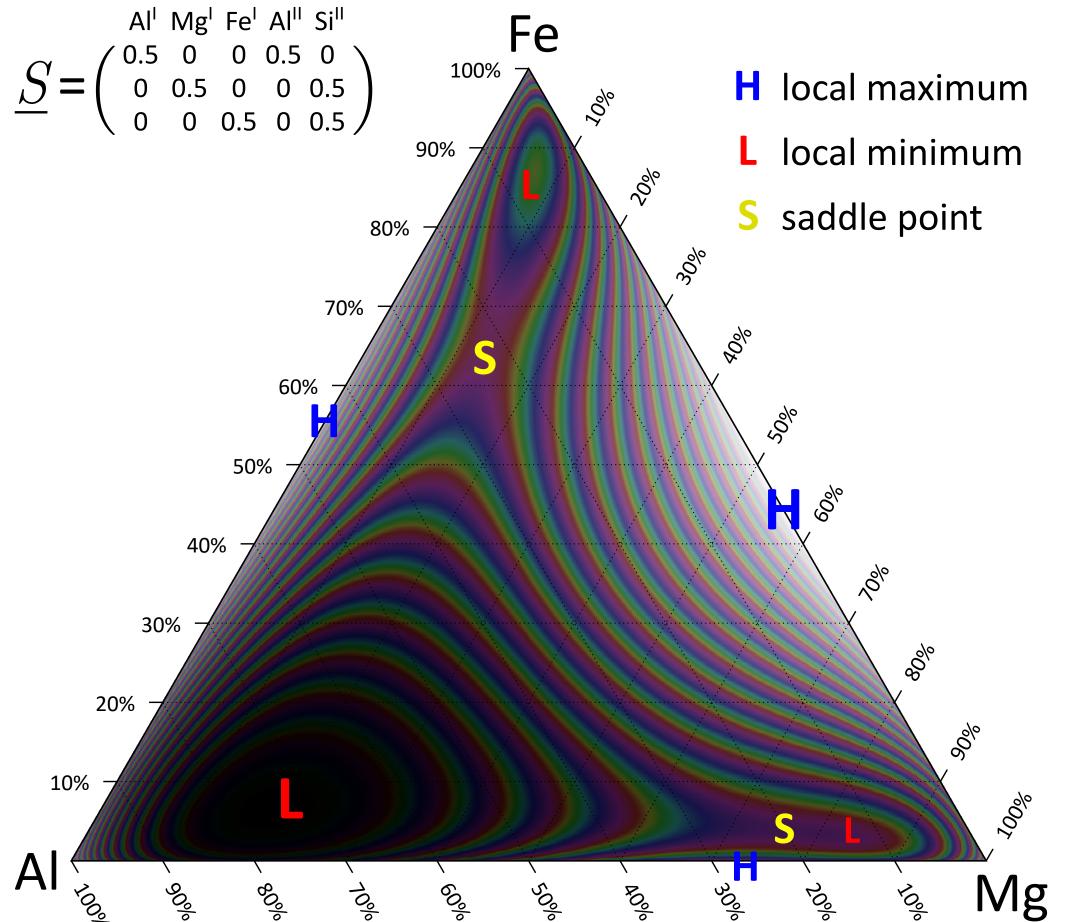
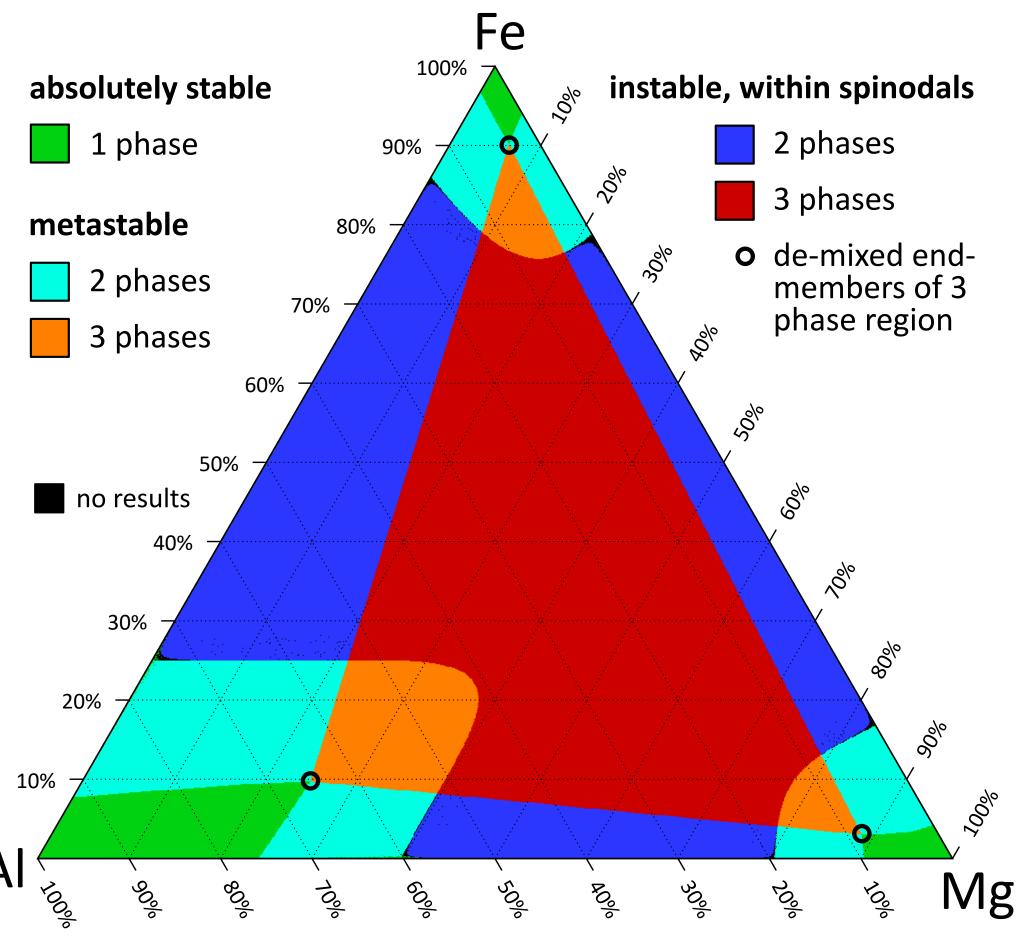
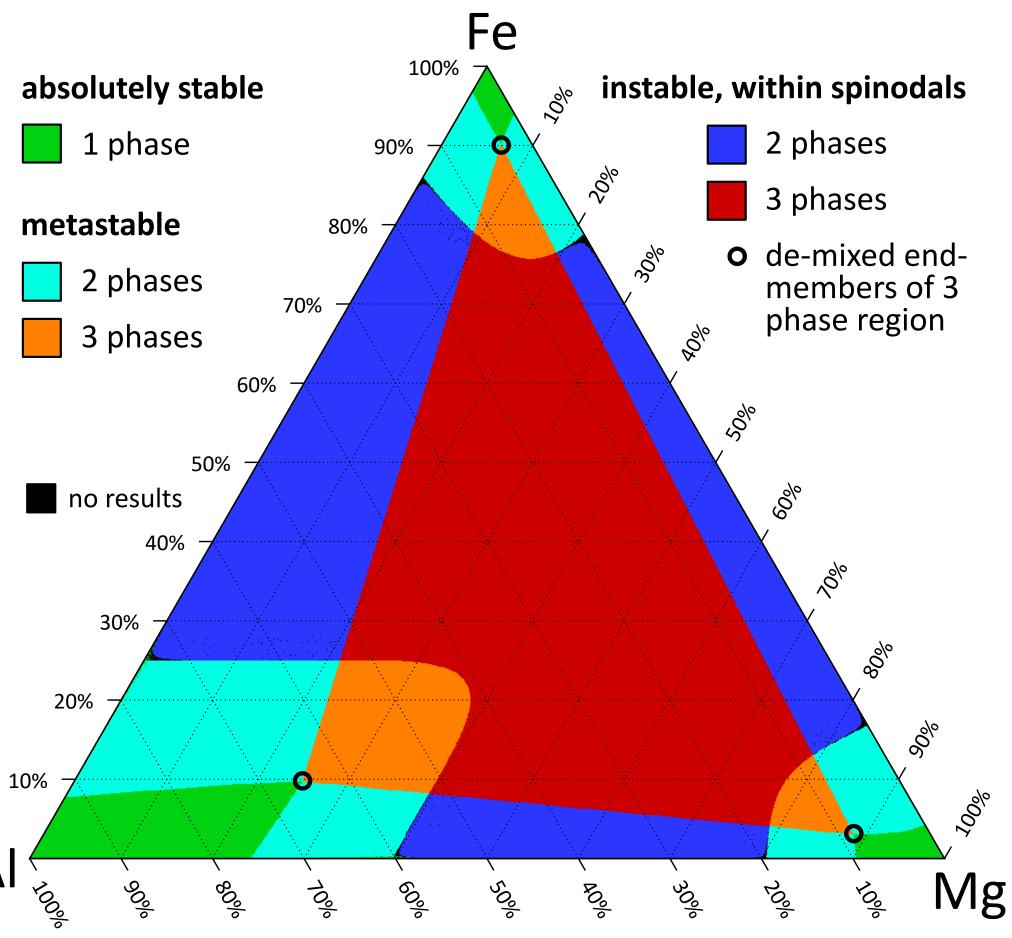
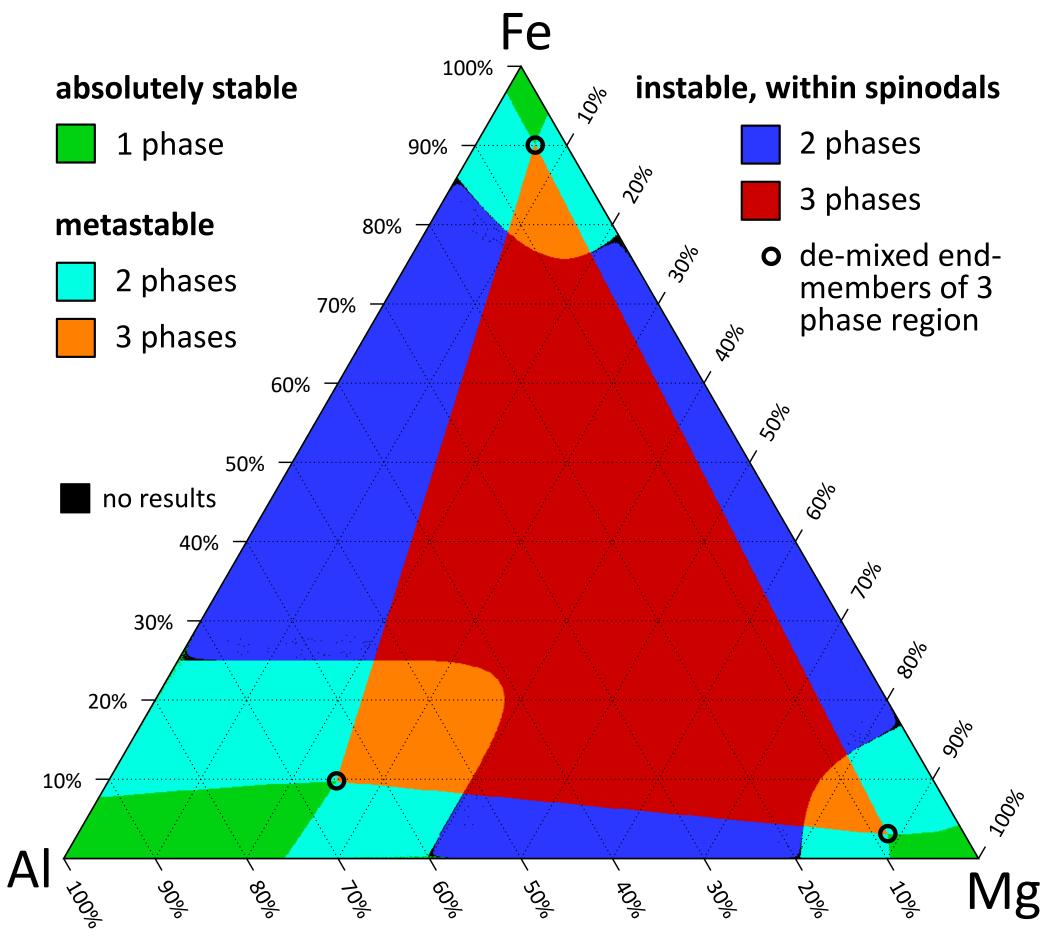


Figure 2: Miscibility landscape. (see text on the right)







G_{mix}

The mixing model

The full expression for the Gibbs energy of mixing G_{mix} of an *n*-phase system based on the Holland and Powell (2011) system is:

$$=\sum_{i=1}^{n} x_{i} \left(G_{i} + \delta G_{i} + RT \ln \prod_{j=1}^{n} \left(\sum_{k=1}^{n} S_{jk} x_{k} \right)^{S_{ji}} \right) + \frac{2}{\sum_{k=1}^{n} a_{k} x_{k}} \sum_{i=1}^{n-1} \sum_{j>i}^{n} x_{i} x_{j} \frac{a_{i} a_{j}}{a_{i} + a_{j}} W_{ij}$$
(1)

 x_i is the normalized ($\Sigma x_i = 1, x_i \ge 0$) molar amount of phase *i*, and G_i its pure end-member Gibbs energy. δG_i summarizes all constant corrections to this while in mixture. Also any parameter may depend on pressure P and temperature T. R is the gas constant.

Ideal mixing on sites (IMOS) is described by an $n_s \times n$ matrix <u>S</u>, where $S_{ii} \ge 0$ is the site occupancy of site j by end-member *i*.

Non-ideal mixing (NIM) is described within the asymmetric formalism (Holland and Powell, 2003) by asymmetry parameters $a_i \ge 0$ and pairwise interaction energies W_{ij} between phases i and j.

For S = 1 the identity matrix and all $W_{ij} = 0$, (1) reduces to the well-known simple ideal mixing equation: $G_{mix} = \sum_{i=1}^{n} x_i (G_i + RT \ln x_i).$

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Miscibility gaps

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Miscibility gaps can occur if, for a given composition x, there is a linear composition of two or more different compositions $\Sigma(b_j \mathbf{x}'_j) = \mathbf{x}$ for which $\Sigma[b_j G_{mix}(\mathbf{x}'_j)] \le G_{mix}(\mathbf{x})$. Spontaneous de-mixing into this combination of phases will occur, when x is within the spinodals of the system. Within the **binodals**, the composition is metastable.

Available TD programs deal with this by using a priori knowledge of the behaviour of the mixing model, limiting their applicability to cases, where such is available. We are working on a semi-analytic al-

gorithm to characterize a mixing model in terms of

Results for the model of Fig. 1 are shown in Fig. 2. Checking for the spinodals can be done analytically by checking the sign of the smallest eigenvalue of $\partial^2 G_{mix}(\mathbf{x})/\partial \mathbf{x}^2$. As the miscibility landscape is completely determined by the second derivatives of G_{mix} , it can be transformed by linear functions without changing its character. Making $\nabla G(\mathbf{x}) = \mathbf{0}$ locally that way is sufficient to determine whether x lies on the lower convex hull (LCH) of G_{mix} and thus is absolutely stable. In similar fashion, up to n supports of the LCH (black circles in Fig. 2) can be found. Compositions within them are known to de-mix into

Our algorithm is currently being optimized and validated, and can be expected to be published in its entirety in due course.